INTERNATIONAL JOURNAL OF ENGINEERING SCIENCES & MANAGEMENT A COMPARATIVE STUDY ON PROPERTIES OF FORGED AND HEAT TREATED D TYPE TOOL STEEL

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ABSTRACT

In this work an attempt has been made to improve the properties of cutting tool like hardness, toughness & wear resistance by forging & heat treatment process. Here we have selected D-Series tool steel (D2,D3) for our project, These tool steels are used in forging dies, die-casting die blocks, drawing dies etc., hence we made an attempt to check whether these tool steel can be used as a cutting tool at high temperature. The experiment evaluates Rockwell hardness test, toughness test & wear test. The result reveals that, hardness of D-series tool steels improved by forging & heat treatment process than it was in raw material case & the heat treated D series-tool can be operated at high temperature without wear.

Keywords: D-type tool steels, forging, heat treatment, hardness, toughness tests

I. INTRODUCTION

D-type tool steels contain between 10% and 18% chromium. These steels retain their hardness up to a temperature of 425 °C (797 °F). Common applications for these tool steels include forging dies, die-casting die blocks, and drawing dies. Due to their high chromium content, certain D-type tool steels are often considered stainless or semi-stainless; however their corrosion resistance is very limited due to the precipitation of the majority of their chromium and carbon constituents as carbides. The heat treatment to which a tool has been subjected has a marked influence on cutting performance of tool steel. The general heat treatment schedules applied to tool steels are shown in figure 1.1.



figure 1. Heat treatment schedules applied to tool steels

Austenitizing is a very critical step in the hardening of tool steel. It is in this step that the final alloy elements are partitioned between the austenitic matrix (which will transform to martensite) and the retained carbides. This partitioning fixes the chemistry, volume fraction, and dispersion of the retained carbides. The retained alloy carbides not only contribute to Wear resistance, but also control austenitic grain size. The finer the carbides and the larger the volume fraction of carbides, the more effectively austenitic grain growth is controlled. If during heating the austenitizing temperature is high, the carbide will dissolve to a large extent, and the precipitation of cementie on cooling will have a greater tendency to take place at coarse austenite grain boundaries. If, however, the carbide has not been completely dissolved and large quantities remain in the form of rounded particles throughout the matrix, carbide precipitation will take place on these preexisting points, and the network of cementite surrounding the grain boundary will not form. Thus, overly high austenitizing temperatures must be avoided so as to prevent grain growth

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which can lead to problems with cracking, retained austenite, and excessive distortion. Relatively slow oil quenching or air cooling for hardening of tool steels can lead to grain boundary carbide formation, which makes tool steel susceptible to intergranular failure. The high harden ability of tool steels effectively suppresses perlite formation at all cooling rates. Bainite formation is also readily suppressed except in heavy sections, which cool slowly. However by slow cooling, the formation of carbides on austenite grain boundaries is difficult to suppress. Small amounts of carbides do not significantly affect hardness but may lower tool steel fracture resistance, leading to quench cracking, and intergranular fracture of tool steels and reduced performance of hot work tool steels. A number of investigations have shown that the presence of the grain boundary carbides significantly reduces toughness of hardened and tempered tool steels. Tool steels hardened in an oxidizing atmosphere scale freely; the D2 and D3 tool steels cannot be hardened in this manner without excessive decarburization. Steels that can be hardened satisfactorily in an oxidizing atmosphere generally have low chromium content (1% or less) and do not require a high hardening temperature (no greater than 870°C). In order to protect the tool steel surface from decarburization and scaling during heat treatment, the furnace medium must be kept neutral. Otherwise decarburization will lead to soft surfaces and cause cracking due to the formation of residual tensile stresses in the surface. A possible explanation of this mechanism is that the reductions in carbon content raise the martensite transformation temperature. Thus, on quenching, the outer layers transform first at a much higher temperature, and when the core transforms and expands, it puts the outer layer in tension.

II. METHOD AND MATERIAL

The raw materials selected are D-series tool steels (D2, D3,). There composition is shown in table 2.1 and 2.2. The selected D2 and D3 tool steel images are shown in figures 2.1 and 2.2. D-series tool steels are selected because, According to American iron and steel institute (AISI), D2 & D3 steel contains high-carbon, high-chromium alloyed with molybdenum, vanadium and characterized by:

- 1. High wear resistance.
- 2. High compressive strength.
- 3. Good through-hardening properties.
- 4. High stability in hardening.
- 5. Good resistance to tempering-back.
- 6. Low cost compare to other tool steels.
- 7. Easily available.

AISI D2 & D3 steels are the most widely used steel in the industry. AISI D2 & D3 steels are recommended for tools requiring very high wear resistance, combined with moderate toughness, these steels are most suitable for testing. Forging process:

D type tool steels D2and D3 are subjected to Press Forging individually by using belt drop hammers of weight 500N shown in figure 2.5. The materials are hot forged by heating it to above itsre-crystallization temperature shown in figure 2.4. In process the materials is heated in furnace to 1000oC shown in figure 2.3. The Dimension of D-type tool steel before forging is 30*30*100 (b * h * l). The Dimension of D-type tool steel after forging is varied 10,20 and 30 percentage shown in figure 2.6.

Heat Treatment process

Heat treatment is a process where the material is subjected to certain temperature and cooling in different ways. Heat treatment involves heating or chilling the material normally to extreme temperatures. It leads to alter the physical properties of a material, this helps to harden or soften the material. There are several Heat treatment techniques, some of them are annealing, Quenching, nitriding, case hardening, precipitation decarburization will lead to soft surfaces and cause cracking due to the formation of residual tensile stresses in the surface. A possible explanation of this mechanism is that the reductions in carbon content raise the martensite transformation temperature. Thus, on quenching, the outer layers transform first at a much higher temperature, and when the core transforms and expands, it puts the outer layer in tension.

Strengthening and tempering . In this work D-type tool steel material is subjected to Heat Treatment by quenching process. It is the process includes rapid cooling of the material after heating to high temperature. Here the materials

are placed in Electrically Heated Coil furnaces as in figure 2.7 and 2.8. Then the material is heated to the prescribed temperature as in figure 2.9. After attaining the prescribed temperature the red heated material as in figure 2.10 and 2.11 is placed in air blower as shown in figures 2.12 and 2.13. Then the material is allowed to cool rapidly as in figure 2.14.

There are several types quenching process they are:

- 1. Air Quenching
- 2. Water Quenching
- 3. Oil Quenching

Air quenching is the heat treatment process used in our project work. Air quenching is heating the tool steel material to high temperature and cooling rapidly by blowing air forcibly. For heat treatment process the below conditions were followed

- 1. Heating temperature 750 to 8000c.
- 2. Type of furnace Electrical induction Coil furnaces.
- 3. Used Air blower Specification Voltage = 220 to 440V. Current = 3.6A Speed = 2800 rpm.
- 4. Cooling time -30min.

III. RESULTS AND DISCUSSION

Hardness test

Table 1: Rockwell hardness of the D-series tool steel.

type of materia l	Raw material	Forging			Forging & heat treatment		
		10 %	20%	30%	10%	20%	30%
D2	60.66	63.66	67	61.5	64.33	61	61.66
D3	63.16	65	60.33	58.66	66.33	60.33	64.16



Figure 2: Hardness variation between forged D2 steel v/s Raw material.



Figure 3: Hardness variation between forged+heat treated D2 steel v/s Raw material.



Figure 4: Hardness variation between forged+heat treated D3 steel v/s Raw material.



Figure 5: Hardness variation between forged D3 steel v/s Raw material.



Figure 6:Impact strength variation between forged D2 steel v/s Raw material.



Figure 7:Impact strength variation between forged D3 steel v/s Raw material.



Figure 8:Impact strength variation between forged D3 steel v/s Raw material.



Fig 9: wear rate of D2 specimen at load 2kg,5mins, at room temperature



Fig 10: wear rate of D2 specimen at load 2kg, 10mins, at room temperature



Fig 11: wear rate of D3 specimen at load 2kg, 5mins, at room temperature



Fig 12: wear rate of D3 specimen at load 2kg, 10mins, at room temperature

IV. CONCLUSION

From the above results it is observed that hardness value is more in heat treated and forged heat treated for all D series tool steels. Hardness is improved in D2 and D3toolsteels at forged + heattreated type. In wear test it is observed that heat treated tools are less worn out compared to others. Also from graph it is observed that at room temperature in both D2 and D3 type Forged +Heat treated tool steels will be less worn out. Future work has to be carried out to concentrate on high temperature wear, toughness and to improve toughness of the material, Along with this a tool has to be made with standard tool nomenclature and tool life has to be calculated

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